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Nb₃Sn Short Sample Calculations and Strain Effects

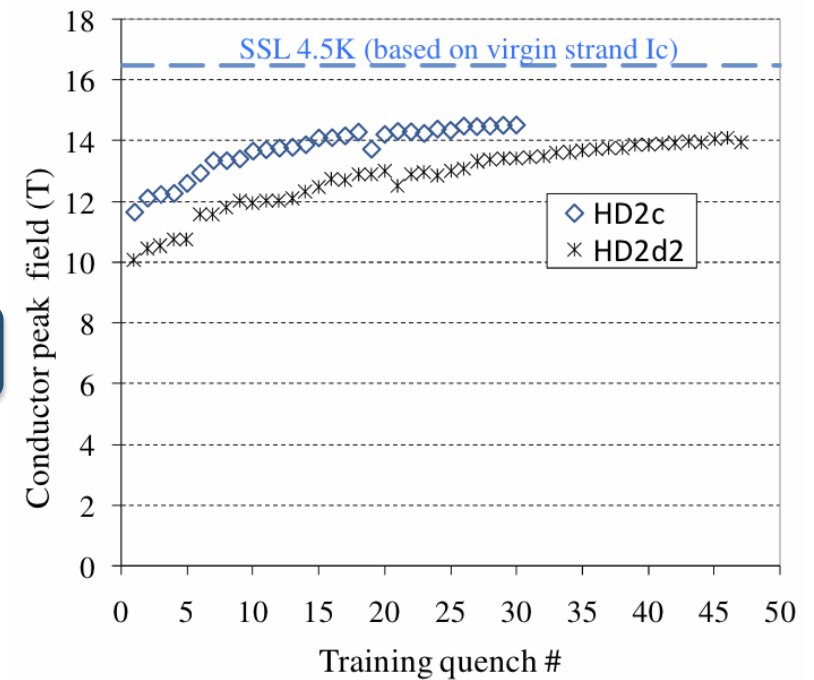
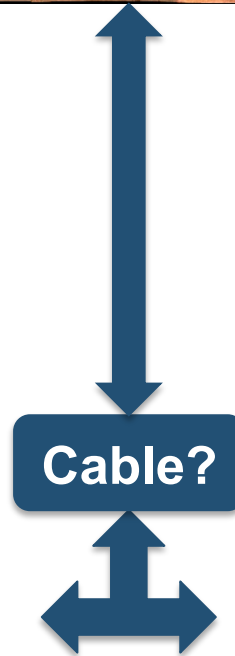
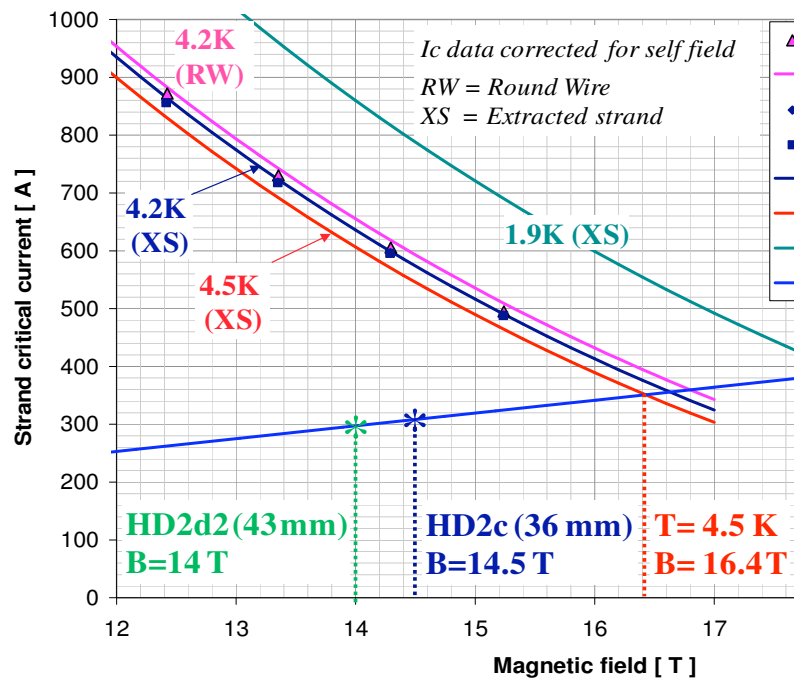
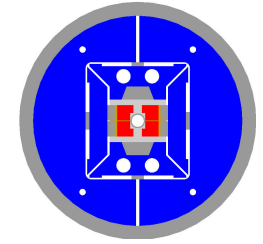
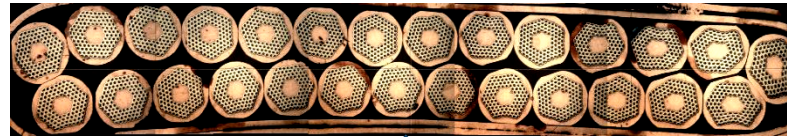
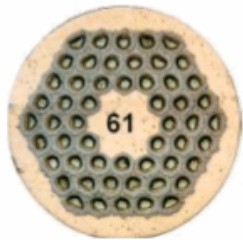
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November 5, 2009 – LARP CM13 Meeting, Port Jefferson, NY

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U.S. Department of Energy under contract No. DE-AC02-05CH11231

Example: LBNL – HD2



Performance is mainly judged from extracted strand data: What is $I_c(B, T, \epsilon)$?

I_c scaling vs. B , T , ε

- Separation of parameters:

$$F_p = J_c(B, T, \varepsilon) \times B$$

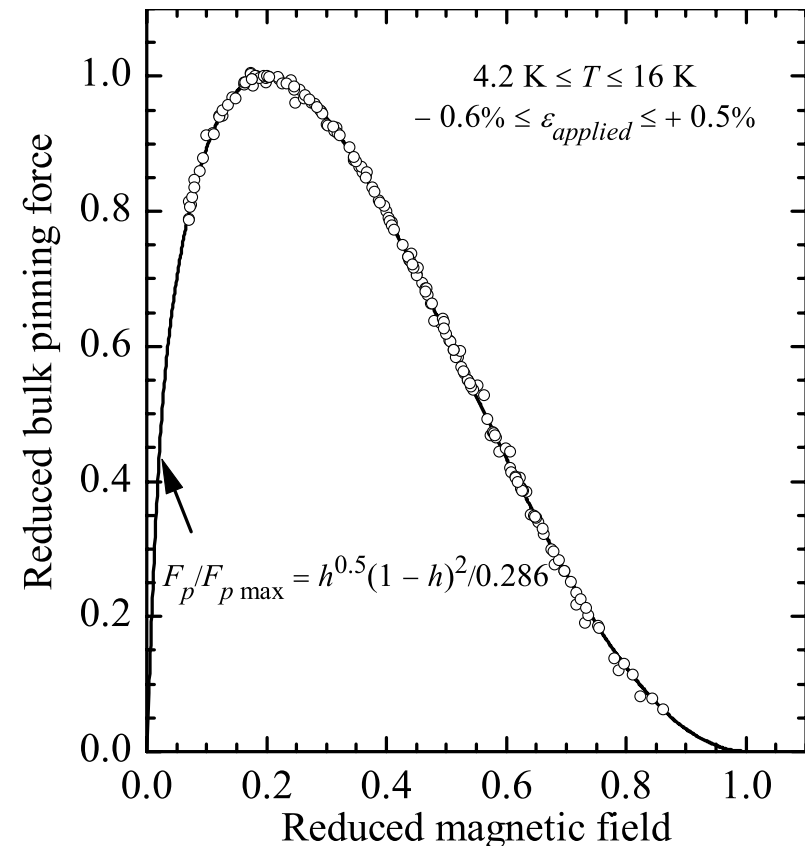
$$= C g(\varepsilon) h(t) f_p(b)$$

- $t = T/T_c$, $b = B/B_{c2}$
 - T_c and B_{c2} are *effective* values
- $g(\varepsilon)$ = some function of strain
 - Common: $g(\varepsilon) = s(\varepsilon) \equiv B_{c2}(\varepsilon)/B_{c2m}$

$$F_p \propto b^{0.5}(1 - b)^2$$

Magnetic field dependence

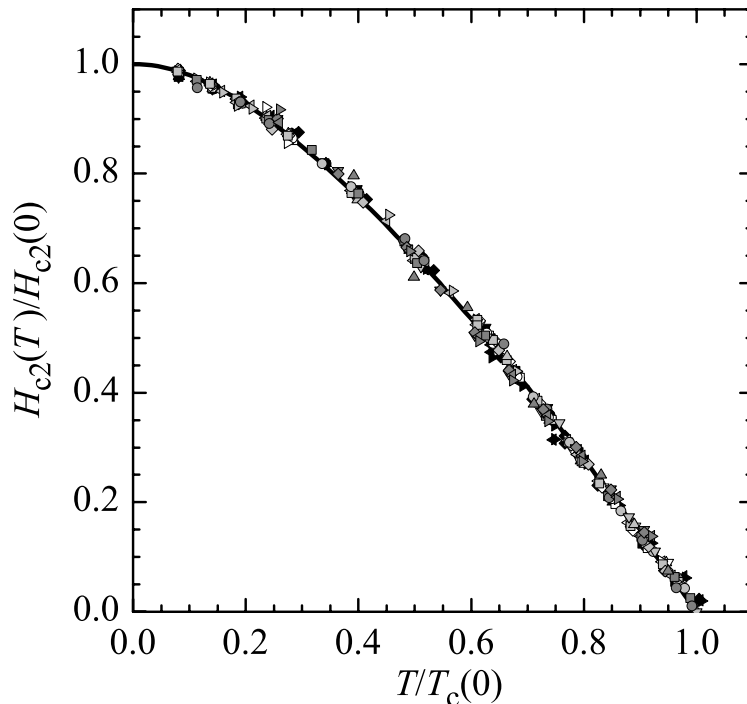
- $F_p = C g(\varepsilon) h(t) f_p(b)$



Godeke et al., *Supercond. Sci. Techn.* **19**, R100 (2006)

Temperature dependence

- $F_p = C g(\varepsilon) h(t) f_p(b)$



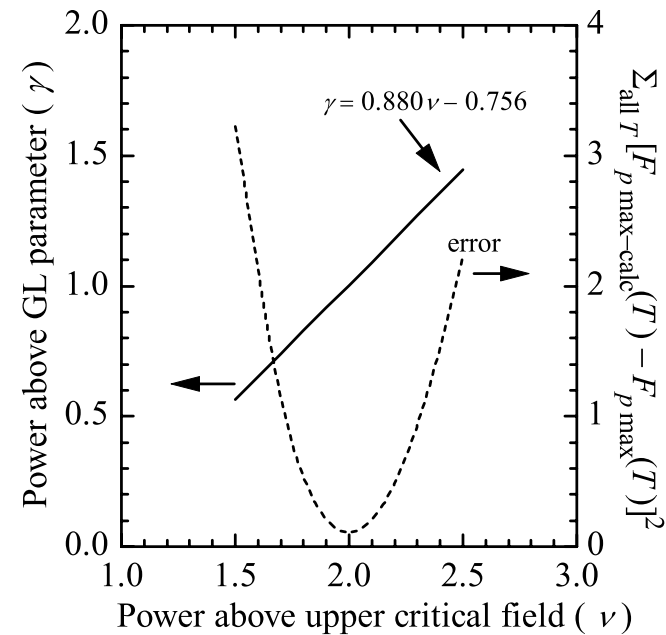
Godeke et al., *J. Appl. Phys.* **97**, 093909 (2005)

- $B_{c2}(t)/B_{c2}(0) = \text{MDG}(t) \approx 1 - t^{1.52}$

$$F_p \propto \text{MDG}(t) (1 - t^2)$$

General form

- $h(t) \propto B_{c2}(t)^\nu / \kappa_1(t)^\nu = B_{c2}(t)^{(\nu - \gamma)} B_c(t)^\gamma$



Godeke et al., *Supercond. Sci. Techn.* **19**, R100 (2006)

- $h(t) \propto B_{c2}(t) B_c(t)$

- $h(t) = \text{MDG}(t)(1 - t^2)$
 $\approx (1 - t^{1.52})(1 - t^2)$

Resulting relations

- $F_p = C \textcolor{red}{g}(\varepsilon) \textcolor{green}{h}(t) \textcolor{blue}{f}_p(b)$
 - $I_c(B, T, \varepsilon) B = C \textcolor{red}{s}(\varepsilon) \textcolor{green}{MDG}(t) (1 - t^2) \textcolor{blue}{b}^{0.5}(1 - b)^2$

Mathematical re-hash: Godeke, Mentink, et al., *IEEE Trans. Appl. Supercond.* **19**, 2610 (2009)

$$I_c(B, T, \varepsilon) = C' (1 - t^2) b^{-0.5}(1 - b)^2$$

with

- $t = T/T_c(0, \varepsilon)$
- $T_c(0, \varepsilon) = T_{cm}(0) s(\varepsilon)^{1/3}$
- $b = B/B_{c2}(T, \varepsilon)$
- $B_{c2}(T, \varepsilon) = B_{c2m}(0) \text{MDG}(t) s(\varepsilon)$
($= B_{c2m}(0) (1 - t^{1.52}) s(\varepsilon)$)

To what extent does the community agree?

Independent, objective comparison of alternatives

Par.

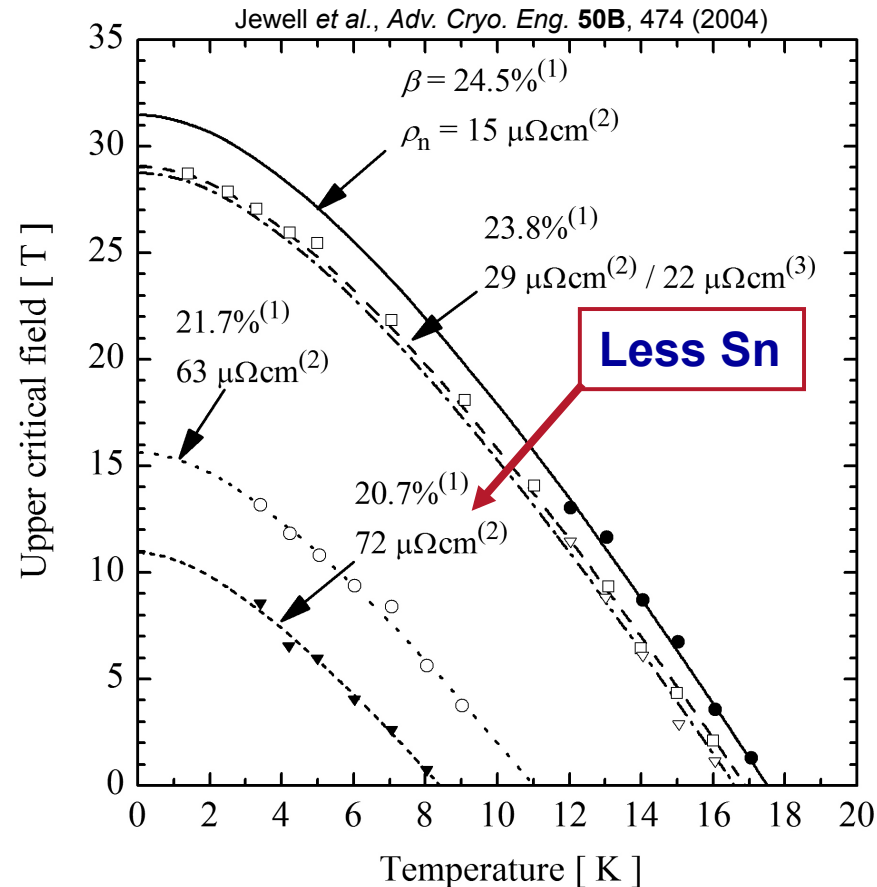
	$g(\varepsilon)$	$h(t)$	$f_I(b)$	$s(\varepsilon)$	
Ekin	$[s(\varepsilon)]^{(1)}$	$(1-t^v)^v$	$b^p(1-b)^q$	$s(\varepsilon) = 1 - a \varepsilon ^{1.7}$	10 – 14
Summers et al.	$s(\varepsilon)$	$[1 - 0.31 t^2 (1 - 1.77 \ln(t))]^{0.5} (1-t^2)^{2.5}$	$b^{0.5}(1-b)^2$	$s(\varepsilon) = 1 - a \varepsilon ^{1.7}$	4
Durham	$[s(\varepsilon)]^{\frac{w(n-2)+2+i}{w}}$	$(1-t^v)^{n-2} (1-t^2)^2$	$b^p(1-b)^q$	$s(\varepsilon) = 1 + c_2\varepsilon^2 + c_3\varepsilon^3 + c_4\varepsilon^4$	13 – 17
Twente.	$s(\varepsilon)$	$(1-t^{1.52})(1-t^2)$	$b^{0.5}(1-b)^2$	$s(\varepsilon) = 1 + \frac{C_{a1} \left(\sqrt{\varepsilon_{sh}^2 + \varepsilon_{0,a}^2} - \sqrt{(\varepsilon - \varepsilon_{sh})^2 + \varepsilon_{0,a}^2} \right) - C_{a2}\varepsilon}{1 - C_{a1}\varepsilon_{0,a}}$ $\varepsilon_{sh} = \frac{C_{a2}\varepsilon_{0,a}}{\sqrt{C_{a1}^2 - C_{a2}^2}}$	7
Markiewicz	-	-		$s(\varepsilon) = \frac{1}{1 + c_2\varepsilon^2 + c_3\varepsilon^3 + c_4\varepsilon^4}$	
Oh and Kim	$[s_n(\varepsilon)]^{2.5} [k(T, \varepsilon)]^{0.5} (1-t^{2.17})^{2.5} (1)$		$b^{0.5}(1-b)^2$	$s_n(\varepsilon) = 1 - \beta \varepsilon ^{1.7} (*)$	9 – 12
ITER-2008	$s(\varepsilon)$	$(1-t^{1.52})(1-t^2)$	$b^p(1-b)^q$	$s(\varepsilon) = 1 + \frac{C_{a1} \left(\sqrt{\varepsilon_{sh}^2 + \varepsilon_{0,a}^2} - \sqrt{(\varepsilon - \varepsilon_{sh})^2 + \varepsilon_{0,a}^2} \right) - C_{a2}\varepsilon}{1 - C_{a1}\varepsilon_{0,a}}$ $\varepsilon_{sh} = \frac{C_{a2}\varepsilon_{0,a}}{\sqrt{C_{a1}^2 - C_{a2}^2}}$	9

Bottura and Bordini., *IEEE Trans. Appl. Supercond.* **19**, 1521 (2009)

Achievable STD across entire space on wires = 3 ÷ 5%

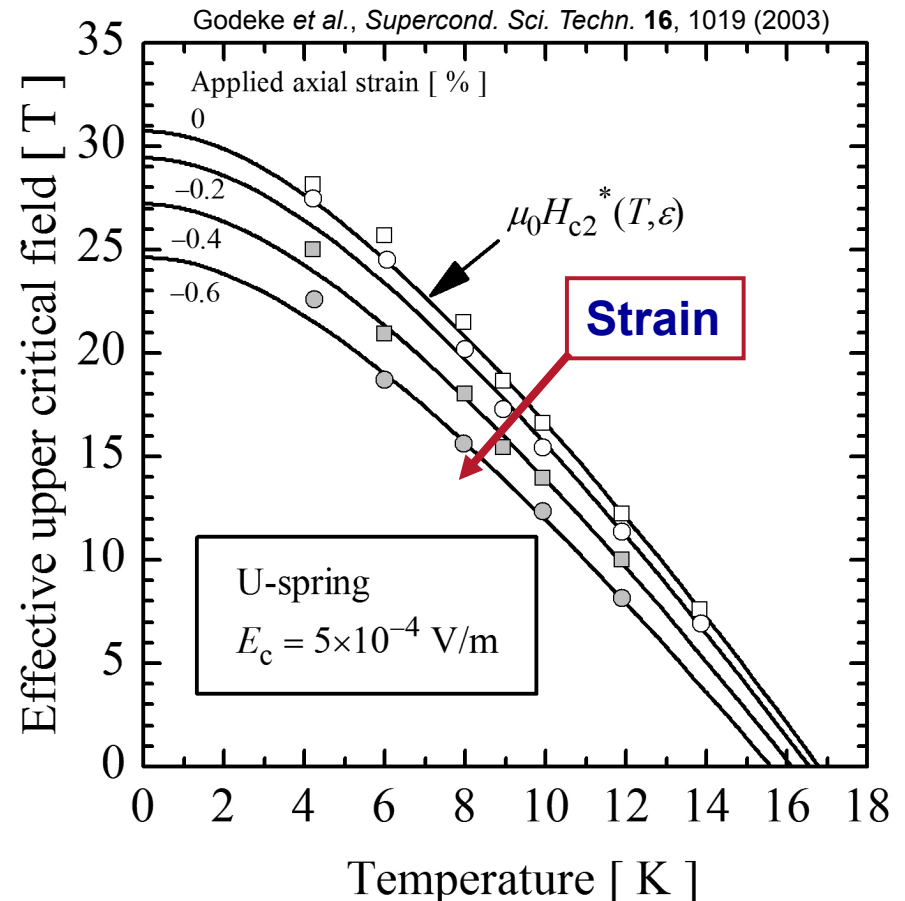
Selection was made. What about the strain function $s(\varepsilon)$?

Composition effects on $B_{c2}(T)$



- Leads to averaged, effective $B_{c2}(T)$

Strain effects on $B_{c2}(T)$



Why does strain affect $B_{c2}(T)$?

Strain modifies

- Lattice vibration modes (phonons)
- Electron-phonon interaction spectrum

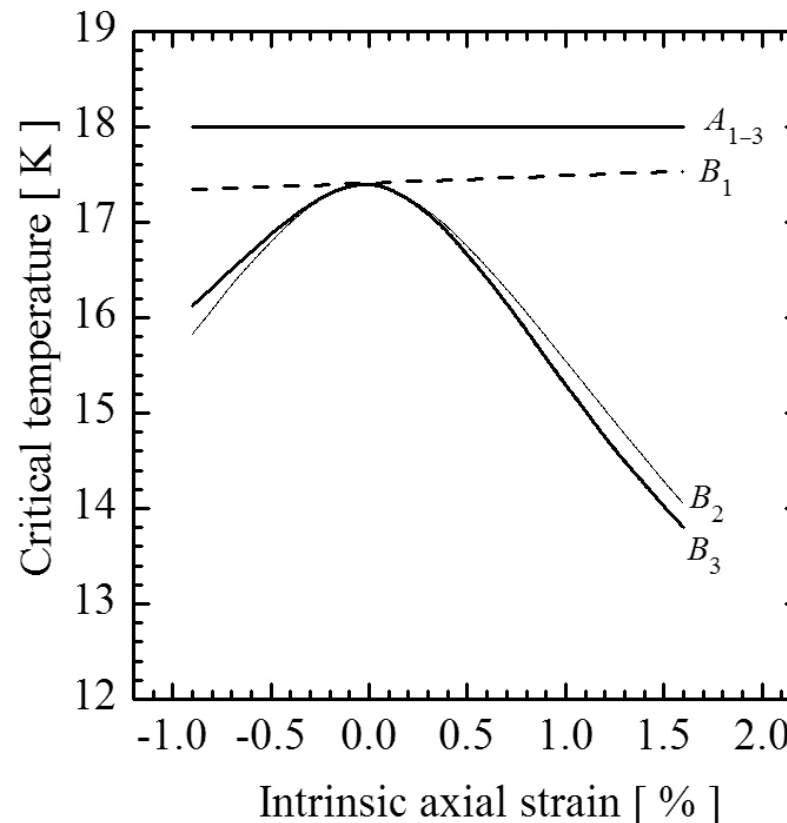
Markiewicz, *Cryogenics* **44**, 676 and 895 (2004)

Markiewicz, *Trans. Appl. Supercond.* **15**, 3368 (2005)

$$\lambda_{\text{ep}} = 2 \int \frac{\alpha^2(\omega) F(\omega)}{\omega} d\omega$$

$$\lambda_{\text{eff}} = \frac{(\lambda_{\text{ep}} - \mu^*)}{(1 + 2\mu^* + 1.5\lambda_{\text{ep}}\mu^* e^{-0.28\lambda_{\text{ep}}})}$$

$$T_c = \frac{0.25 \langle \omega^2 \rangle^{\frac{1}{2}}}{(e^{2/\lambda_{\text{eff}}} - 1)^{\frac{1}{2}}} \quad \mu_0 H_{c2} = \dots?$$



Strain free value
Hydrostatic (1st inv)

Symmetric (2nd inv)
non-hydrostatic

Asymmetric (3rd inv)
non-hydrostatic

Can this be simplified while retaining physics and 3D?

From strain energy function

- All invariants

$$\frac{H_{c2}^*(\epsilon)}{H_{c2}^*(\epsilon = 0)} = (1 - C_h I_1) \left(1 - C_{d,1} \sqrt{J_2} - C_{d,2} J_3 \right)$$

In axial form for wires:

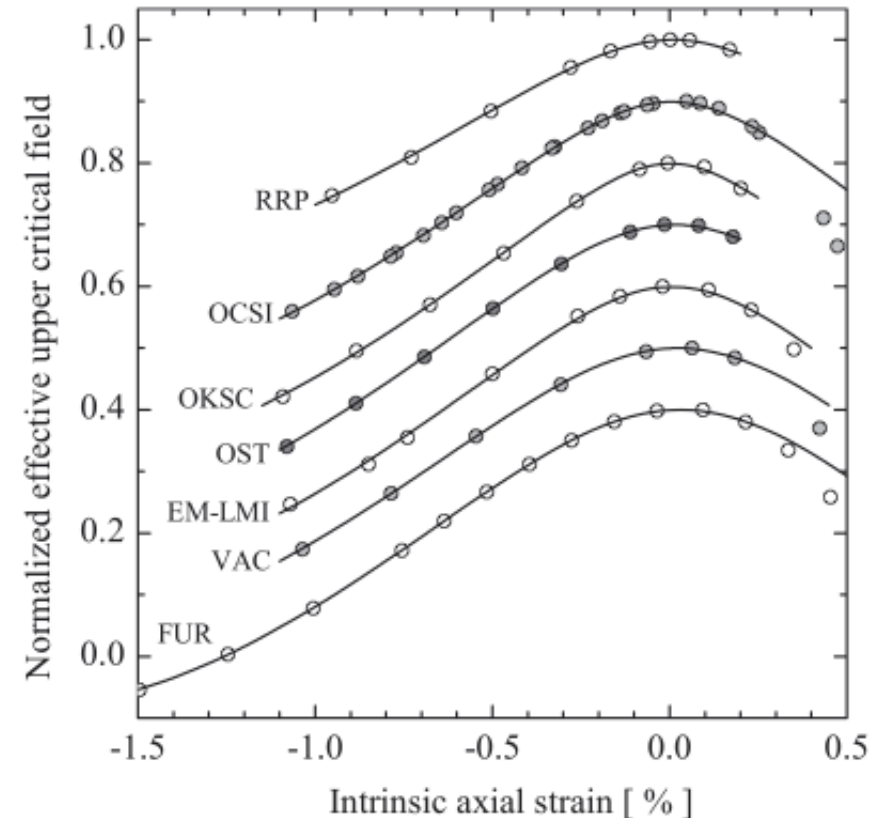
$$\begin{aligned} s(\epsilon_I) &= \frac{H_{c2}^*(\epsilon_I)}{H_{c2}^*(\epsilon_I = 0)} \\ &= \frac{1 - C_{a,1} \sqrt{(\epsilon_I)^2 + (\epsilon_{0,a})^2} - C_{a,2} ((\epsilon_I)^3 - 3(\epsilon_{0,a})^2 \epsilon_I)}{1 - C_{a,1} \epsilon_{0,a}} \end{aligned}$$

- Without loss of accuracy:

- $C_{a2} = 1034 \times C_{a1}$
- 3 fit parameters: C_{a1} , $\epsilon_{0,a}$, ϵ_m
- 3D form to be validated

Comparison to measurement

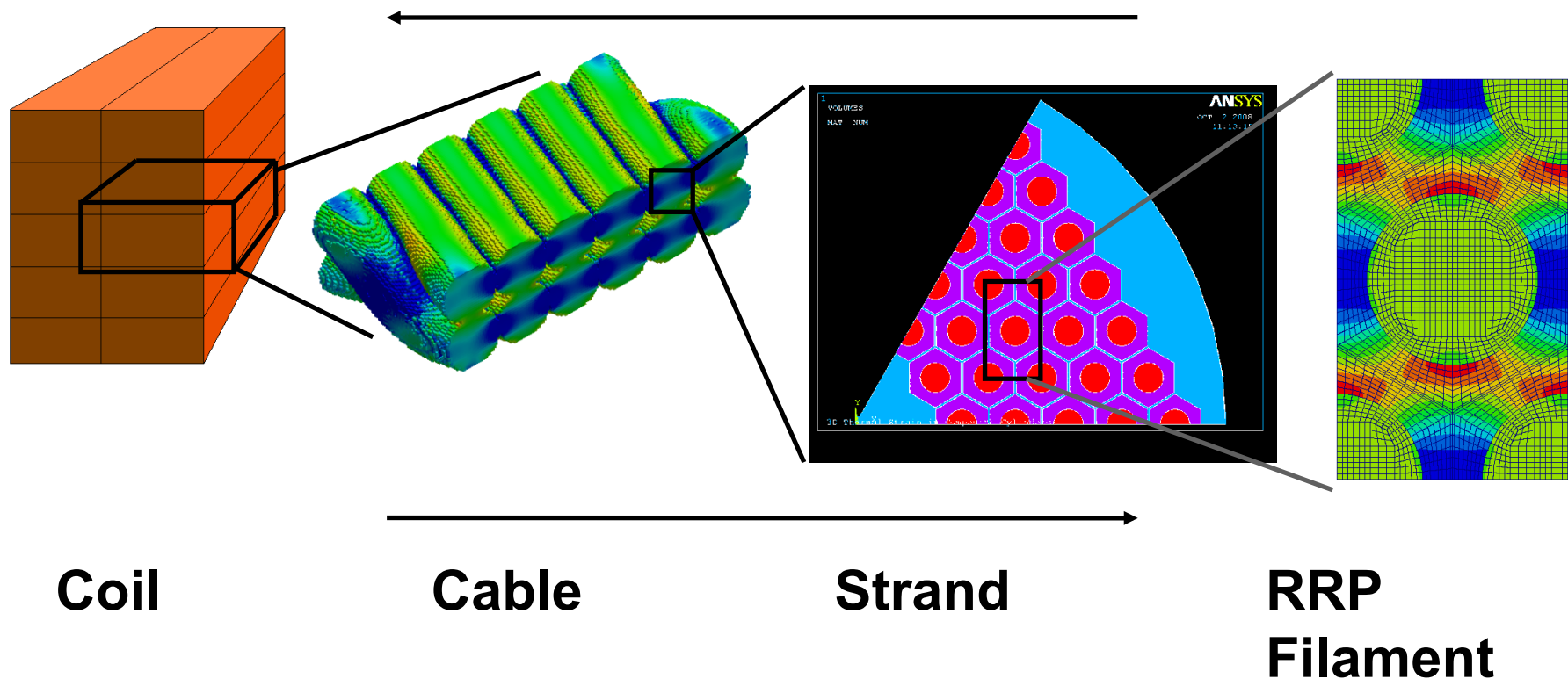
Arbelaez et al., *Supercond. Sci. Techn.* **22**, 025005 (2009)



Why the emphasis on 3D?

3D strain state at the filaments for applied macro-scale loads

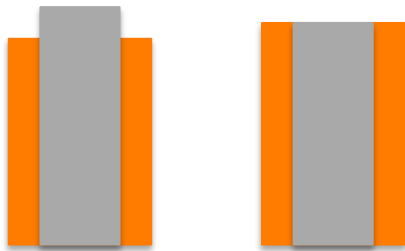
- Arbelaez *et al.*, EUCAS 2009



But, for now, only 1D is feasible

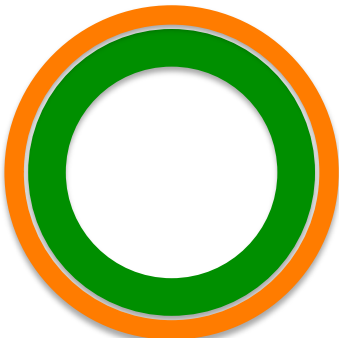
Ballpark axial strain states in Nb₃Sn fractions

Wire: Nb₃Sn in Cu



$$\epsilon_a = -0.35\%$$

Wire on ITER barrel



Ti-6Al-4V: $\epsilon_a = -0.2\%$

SS: $\epsilon_a = -0.3\%$

Ghosh, *BNL report MDN-657-39* (2009)
THERMAL CONTRACTION FROM R.T. TO 4.2 K (OR 10 K)

Copper:	-0.30%
Stainless Steel	-0.27%
Ti-6Al-4V	-0.15%
G-10	-0.28%
Nb ₃ Sn	-0.15%
Inconel 600	-0.27%
Composite Nb ₃ Sn strand:	-0.15% to -0.29%

Iron

-0.2%

Ghosh, *BNL report MDN-657-39* (2009)

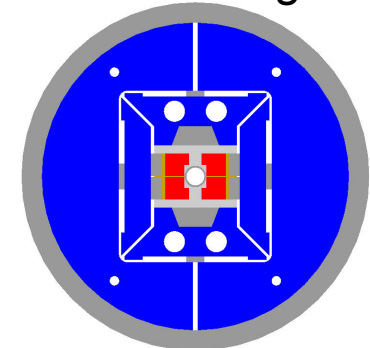
Barrel Material	$\Delta l_c/l_c$	$\Delta \epsilon$
Ti-Al-V	0.00	0
G-10	0.10	-0.10%
SS-304	0.08	-0.07%
SS- Soldered	0.19	-0.15%

Cable in SS holder



$$\epsilon_a = -0.3\%$$

Cable in magnet



Loaded iron enclosure
with Ti-6Al-4V poles

$$\epsilon_a = -0.2\% ?$$

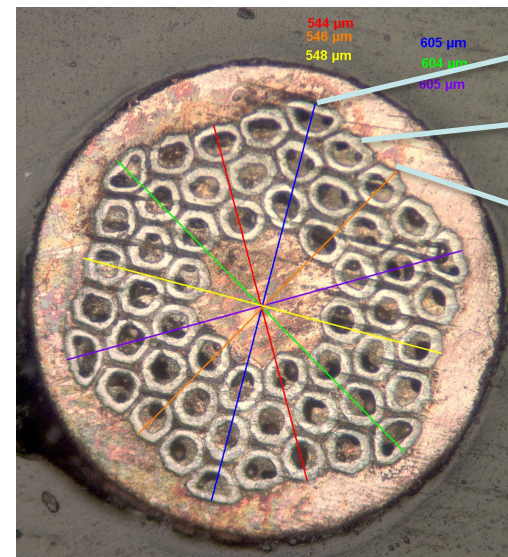
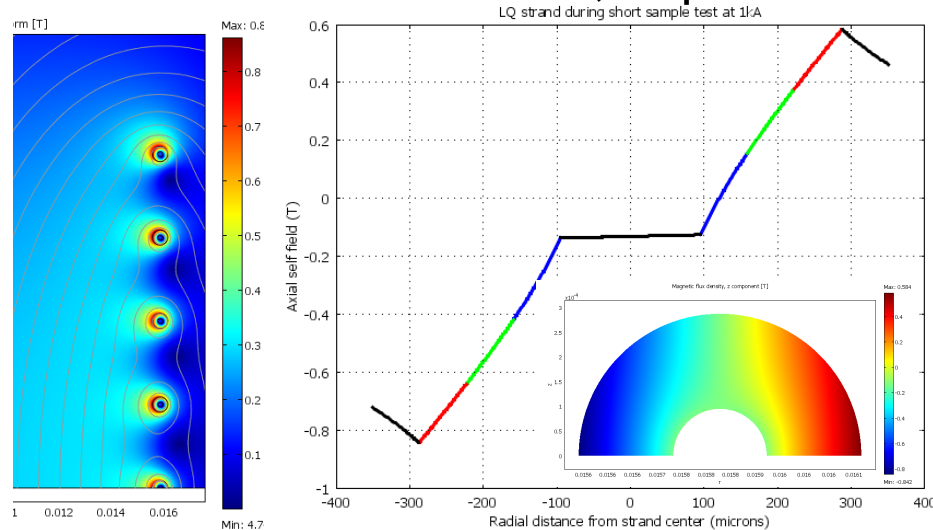
I_c results for extracted strands on Ti-alloy barrels

- Self-field correction required

–Why?

Applied field [T]	Measured I_c [A]	Total field [T]
12.0	571	12.32
11.0	693	11.39
10.0	836	10.47

–How much? Kashikhin, unpublished



0.584 mT/A
0.555 mT/A

0.483 mT/A

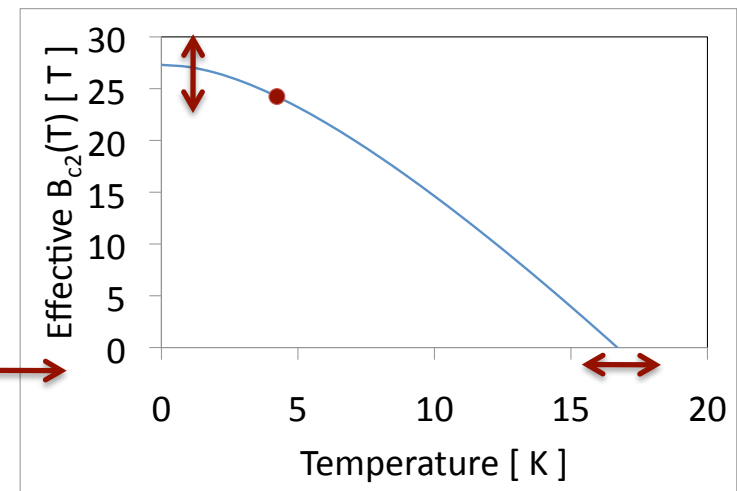
Bapplied [T]	Btotal [T]	Ic-meas	Ic-calc
11.5	11.85	628	625
11.0	11.39	693	691
10.5	10.92	761	762
10.0	10.47	836	838
9.5	10.01	916	919
9.0	9.56	1003	1007
8.0	8.67	1205	1199

Ti-6Al-4V barrel

$\epsilon_a = -0.2\%$

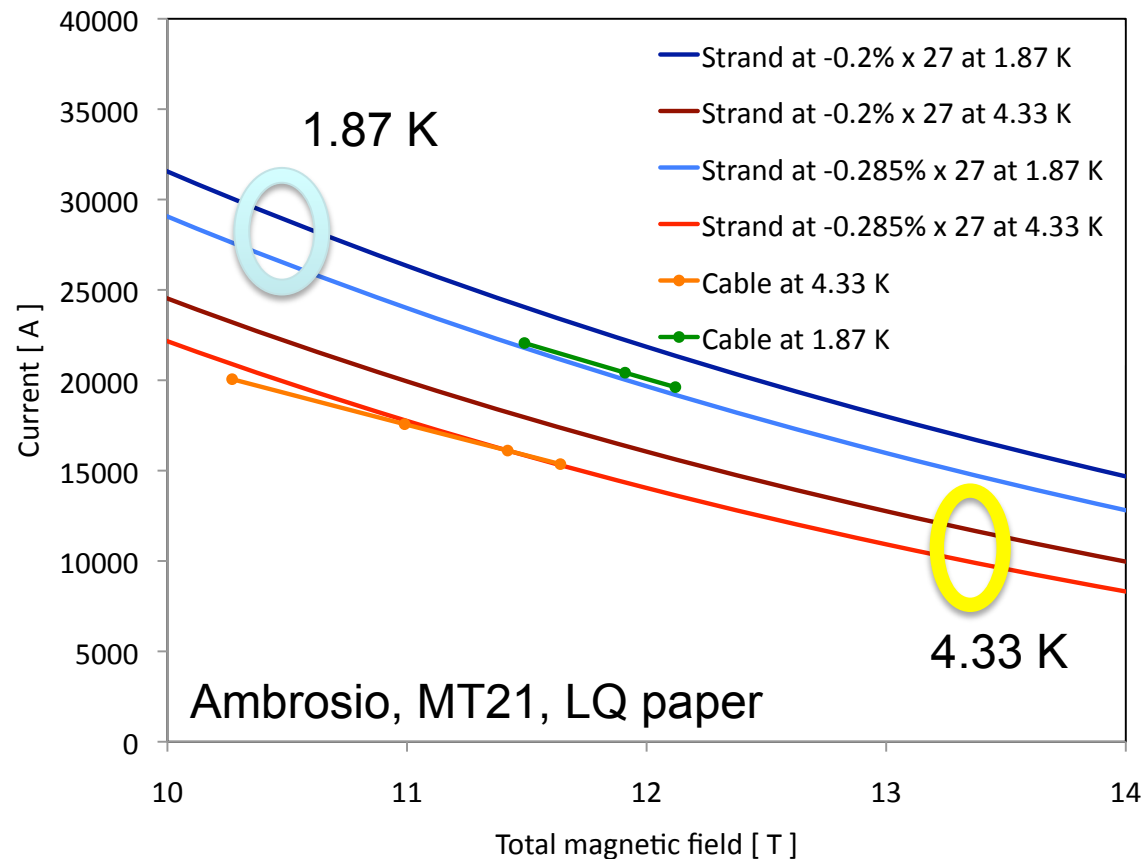
$T = 4.23 \text{ K}$

Parameter	Value	Determined
$C_{a1} [\text{T}]$	48.6	From strain data
$\epsilon_{0,a} [\%]$	0.222	From strain data
$B_{c2m}(0) [\text{T}]$	27.3	Fitted to data
$T_{cm}(0) [\text{K}]$	16.7	Chosen if unknown
$C [\text{AT}]$	54809	Fitted to data



Comparison of strand scaling and SF corrected cable data

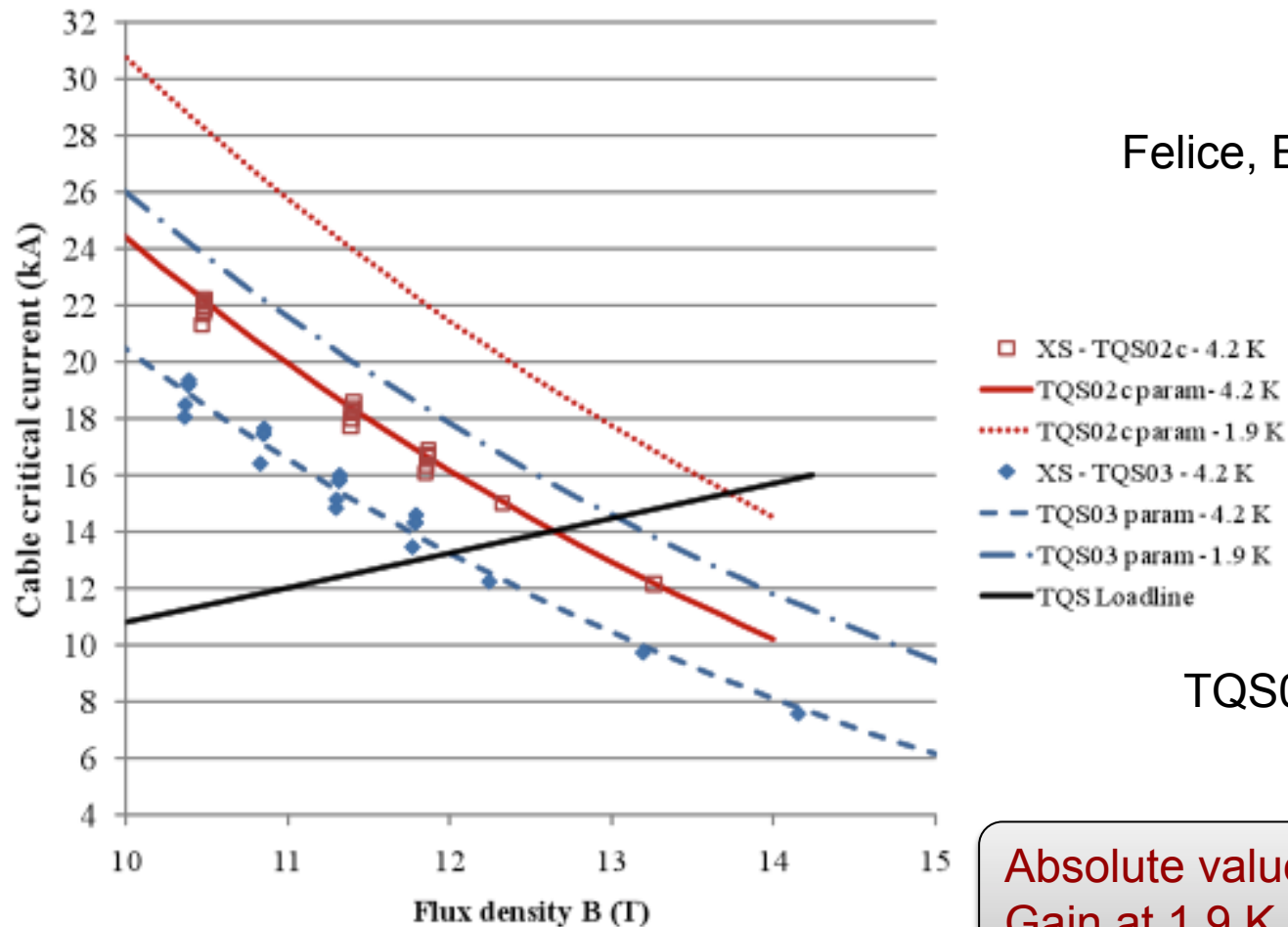
- Cable has additional – 0.085% axial strain compared to XS



What is the SS for the magnet? Closest strain match (barrel?)

Coil performance compared to XS parameterization

- XS scaling based on 4.2 K barrel data and estimated $T_{cm}(0) = 16.7$ K



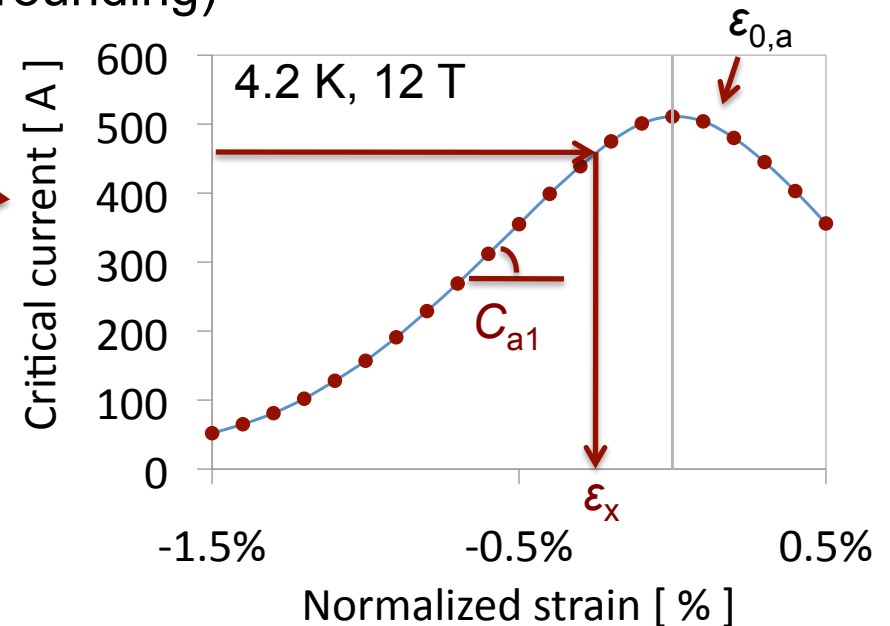
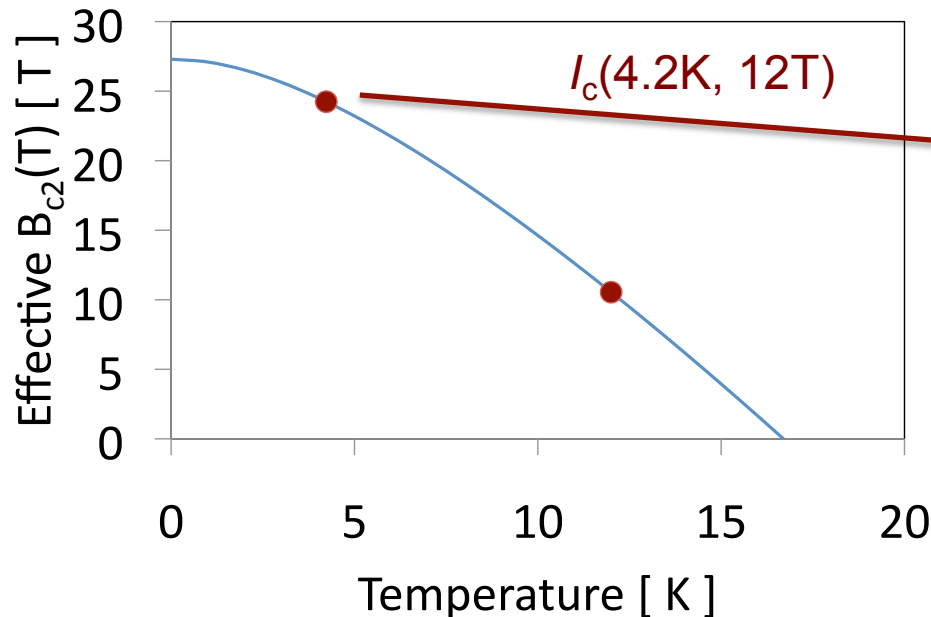
Felice, EUCAS 2009 TQ paper

TQS03a = 93% at 4.3 K
= 93% at 1.9 K

Absolute value SS unknown (strain)
Gain at 1.9 K is accurately *predicted*

What is the minimum data required to fully parameterize an XS?

- $I_c(B)$ at 4.2 and, say, 12 K
 - Provides $B_{c2}(4.2 \text{ K})$ and $B_{c2}(12 \text{ K})$
 - ...and thus $B_{c2}(0, \epsilon_x)$ and $T_c(0, \epsilon_x)$
- $I_c(\epsilon)$ at, say, 12 T and 4.2 K or 12 K
 - Provides ϵ_x , C_{a1} (slope), and $\epsilon_{0,a}$ (peak rounding)



Only limited data is needed.



Summary

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Emerging consensus on scaling relations

- Powers of temperature dependence still argued
- 'Latest greatest' 3D strain model needs 3D verification
- All is scaled axially, due to unknown 3D strain state of Nb_3Sn

Models can reasonably explain differences XS – cable – magnet

- Different strain state cable vs. XS very plausible
- 3D mechanical modeling lacking
- Limited SS measurements required to map $I_c(B, T, \epsilon)$
- Absolute prediction (why 93%?) remains inaccessible (3D strain)
- Relative changes can be truly predictive